

# Spectroscopic Properties of Ho<sup>3+</sup> Doped Zinc Lithium Bismuth Borate Glasses

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**Abstract:** Zinc lithium bismuth borate glasses containing Ho<sup>3+</sup> in (25- x): Bi<sub>2</sub>O<sub>3</sub>:20Li<sub>2</sub>O:20ZnO: 35B<sub>2</sub>O<sub>3</sub>:xHo<sub>2</sub>O<sub>3</sub> (where x=1, 1.5,2 mol %) have been prepared by melt-quenching method. The amorphous nature of the glasses was confirmed by x-ray diffraction studies. Optical absorption spectra were recorded at room temperature for all glass samples. The experimental oscillator strengths were calculated from the area under the absorption bands. Slater-Condon parameter F<sub>2</sub>, Lande's parameter  $\xi_{4f}$ , Nephelauxetic ratio ( $\beta'$ ) and Bonding parameter ( $b^{1/2}$ ) have been computed. Using these parameters energies and intensities of these bands has been calculated. Judd-Ofelt intensity parameters  $\Omega_{\lambda}$  ( $\lambda=2, 4, 6$ ) are evaluated from the intensities of various absorption bands of optical absorption spectra. Using these intensity parameters various radiative properties like spontaneous emission probability, branching ratio, radiative life time and stimulated emission cross-section of various emission lines have been evaluated.

**Keywords:** ZnLiBiB Glasses, Optical Properties, Judd-Ofelt Theory, Rare earth ions.

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## I. Introduction

Zinc lithium bismuth borate glasses find a wide range of technological applications as electro-chemical devices as ionic conductors, optoelectronic devices [1-3]. The glasses containing rare earth in various forms such as network formers, modifiers or luminescent ions are of great deal of interest for their unique optical, electrical and magnetic properties [4]. In order to improve the glass quality and its optical performance a divalent oxide such as ZnO has been added separately beside the other property improving network modifier (NWF) namely Li<sub>2</sub>O.

Among RE<sup>3+</sup> ions, Ho<sup>3+</sup> is an interesting ion for spectroscopic studies, because it exhibits several electronic transitions in the UV and VIS. Glasses containing heavy metal oxides exhibits good non-linear optical properties and good chemical durability [5, 6]. The past literature shows that the rare earth ions find more important application in the preparation of the laser materials [7-10].

In this work, we have studied on the absorption and emission properties of Ho<sup>3+</sup> doped zinc lithium bismuth borate glasses. The Judd-Ofelt theory has been applied to compute the intensity parameters  $\Omega_{\lambda}$  ( $\lambda=2, 4, 6$ ), which are sensitive to the environment of rare earth ion. From these parameters, important optical properties such as radiative transition probability for spontaneous emission, radiative lifetime of the excited states and branching ratio can be estimated.

## II. Experimental Techniques

### Preparation of glasses

The following Ho<sup>3+</sup> doped zinc lithium bismuth borate glass samples (25-x) Bi<sub>2</sub>O<sub>3</sub>:20Li<sub>2</sub>O:20ZnO: 35 B<sub>2</sub>O<sub>3</sub>: xHo<sub>2</sub>O<sub>3</sub> (where x=1, 1.5,2) have been prepared by melt-quenching method. Analytical reagent grade chemical used in the present study consist of Bi<sub>2</sub>O<sub>3</sub>, Li<sub>2</sub>O, ZnO, and B<sub>2</sub>O<sub>3</sub> and Ho<sub>2</sub>O<sub>3</sub>. All weighed chemicals were powdered by using an Agate pestle mortar and mixed thoroughly before each batch (10g) was melted in alumina crucibles in silicon carbide based an electrical furnace.

Silicon Carbide Muffle furnace was heated to working temperature of 1050<sup>0</sup>C, for preparation of Zinc Lithium Bismuth Borate glasses, for two hours to ensure the melt to be free from gases. The melt was stirred several times to ensure homogeneity. For quenching, the melt was quickly poured on the steel plate & was immediately inserted in the muffle furnace for annealing. The steel plate was preheated to 100<sup>0</sup>C. While pouring; the temperature of crucible was also maintained to prevent crystallization. And annealed at temperature of 350<sup>0</sup>C for 2h to remove thermal strains and stresses. Every time fine powder of cerium oxide was used for polishing the samples. The glass samples so prepared were of good optical quality and were transparent. The chemical compositions of the glasses with the name of samples are summarized in **Table 1**

**Table 1** Chemical composition of the glasses

Sample	Glass composition (mol %)
ZnLiBiB (UD)	25 Bi <sub>2</sub> O <sub>3</sub> :20Li <sub>2</sub> O:20ZnO: 35 B <sub>2</sub> O <sub>3</sub>
ZnLiBiB (HO1)	24 Bi <sub>2</sub> O <sub>3</sub> :20Li <sub>2</sub> O:20ZnO: 35 B <sub>2</sub> O <sub>3</sub> : 1 Ho <sub>2</sub> O <sub>3</sub>
ZnLiBiB (HO1.5)	23.5 Bi <sub>2</sub> O <sub>3</sub> :20Li <sub>2</sub> O:20ZnO: 35 B <sub>2</sub> O <sub>3</sub> : 1.5 Ho <sub>2</sub> O <sub>3</sub>
ZnLiBiB (HO2)	23 Bi <sub>2</sub> O <sub>3</sub> :20Li <sub>2</sub> O:20ZnO: 35 B <sub>2</sub> O <sub>3</sub> : 2 Ho <sub>2</sub> O <sub>3</sub>

ZnLiBiB (UD)—Represents undoped Zinc Lithium Bismuth Borate glass specimens

ZnLiBiB (HO) -Represents Ho<sup>3+</sup> doped Zinc Lithium Bismuth Borate glass specimens

### III. Theory

#### 3.1 Oscillator Strength

The intensity of spectral lines are expressed in terms of oscillator strengths using the relation [11].

$$f_{\text{expt.}} = 4.318 \times 10^{-9} \int \epsilon(\nu) d\nu \quad (1)$$

where,  $\epsilon(\nu)$  is molar absorption coefficient at a given energy  $\nu$  (cm<sup>-1</sup>), to be evaluated from Beer–Lambert law. Under Gaussian Approximation, using Beer–Lambert law, the observed oscillator strengths of the absorption bands have been experimentally calculated, using the modified relation [12].

$$P_m = 4.6 \times 10^{-9} \times \frac{1}{cl} \log \frac{I_0}{I} \times \Delta\nu_{1/2} \quad (2)$$

where  $c$  is the molar concentration of the absorbing ion per unit volume,  $l$  is the optical path length,  $\log I_0/I$  is absorptivity or optical density and  $\Delta\nu_{1/2}$  is half band width.

#### 3.2. Judd-Ofelt Intensity Parameters

According to Judd [13] and Ofelt [14] theory, independently derived expression for the oscillator strength of the induced forced electric dipole transitions between an initial  $J$  manifold  $|4f^N(S, L) J\rangle$  level and the terminal  $J'$  manifold  $|4f^N(S', L') J'\rangle$  is given by:

$$\frac{8\pi^2 mc \nu}{3h(2J+1)n} \left| \frac{1}{9} \left[ \frac{(n^2+2)^2}{9} \right] \right| \times S(J, J') \quad (3)$$

where, the line strength  $S(J, J')$  is given by the equation

$$S(J, J') = e^2 \sum_{\lambda=2,4,6} \Omega_{\lambda} \langle 4f^N(S, L) J \| U^{(\lambda)} \| 4f^N(S', L') J' \rangle^2 \quad (4)$$

In the above equation  $m$  is the mass of an electron,  $c$  is the velocity of light,  $\nu$  is the wave number of the transition,  $h$  is Planck's constant,  $n$  is the refractive index,  $J$  and  $J'$  are the total angular momentum of the initial and final level respectively,  $\Omega_{\lambda}$  ( $\lambda = 2, 4, 6$ ) are known as Judd-Ofelt intensity parameters which contain the effect of the odd-symmetry crystal field terms, radial integrals and energy denominators.  $\| U^{(\lambda)} \|^2$  are the matrix elements of the doubly reduced unit tensor operator calculated in intermediate coupling approximation.  $\Omega_{\lambda}$  parameter can be obtained from least square fitting method [15]. The matrix element  $\| U^{(\lambda)} \|^2$  that are insensitive to the environment of rare earth ions were taken from the literature [16].

#### 3.3. Radiative Properties

The  $\Omega_{\lambda}$  parameters obtained using the absorption spectral results have been used to predict radiative properties such as spontaneous emission probability ( $A$ ) and radiative life time ( $\tau_R$ ), and laser parameters like fluorescence branching ratio ( $\beta_R$ ) and stimulated emission cross section ( $\sigma_p$ ).

The spontaneous emission probability from initial manifold  $|4f^N(S', L') J'\rangle$  to a final manifold  $|4f^N(S, L) J\rangle$  is given by:

$$A[(S', L') J'; (S, L) J] = \frac{64 \pi^2 \nu^3}{3h(2J'+1)} \left[ \frac{n(n^2+2)^2}{9} \right] \times S(J', \bar{J}) \quad (5)$$

$$\text{Where, } S(J', J) = e^2 [\Omega_2 \| U^{(2)} \|^2 + \Omega_4 \| U^{(4)} \|^2 + \Omega_6 \| U^{(6)} \|^2]$$

The fluorescence branching ratio for the transitions originating from a specific initial manifold  $|4f^N(S', L') J'\rangle$  to a final many fold  $|4f^N(S, L) J\rangle$  is given by

$$\beta[(S', L') J'; (S, L) J] = \sum_{S, L, J} \frac{A[(S', L) J]}{A[(S', L) J'(\bar{S}, \bar{L})]} \quad (6)$$

where, the sum is over all terminal manifolds.

The radiative life time is given by

$$\tau_{\text{rad}} = \sum_{S' L' J'; S L J} A[(S', L') J'; (S, L) J] = A_{\text{Total}}^{-1} \quad (7)$$

where, the sum is over all possible terminal manifolds. The stimulated emission cross-section for a transition from an initial manifold  $|4f^N (S', L') J' \rangle$  to a final manifold  $|4f^N (S, L) J \rangle$  is expressed as

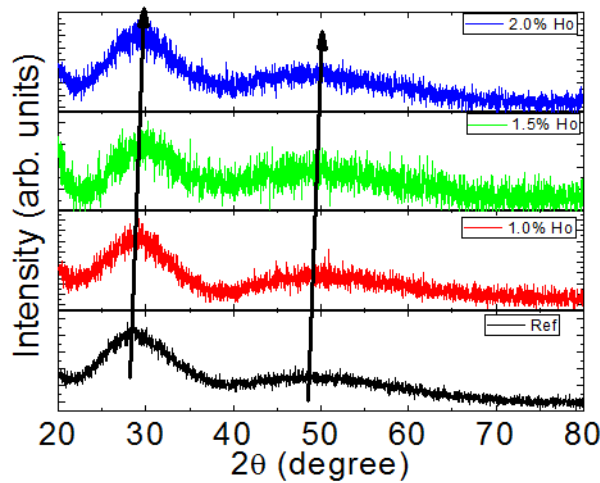
$$\sigma_p(\lambda_p) = \left[ \frac{\lambda_p^4}{8\pi c n^2 \Delta\lambda_{\text{eff}}} \right] \times A[(S', L') J'; (\bar{S}, \bar{L}) J] \quad (8)$$

where,  $\lambda_p$  the peak fluorescence wavelength of the emission band and  $\Delta\lambda_{\text{eff}}$  is the effective fluorescence line width.

## IV. Result And Discussion

### 4.1. XRD Measurement

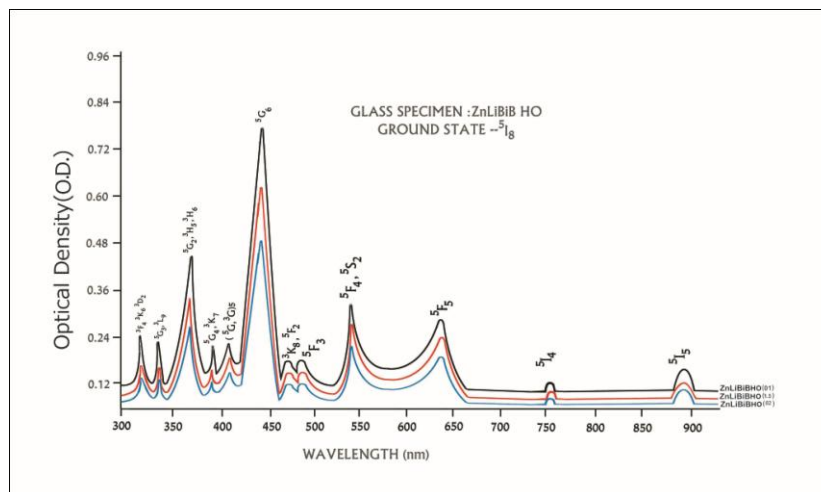
Figure 1 presents the XRD pattern of the samples containing show no sharp Bragg's peak, but only a broad diffuse hump around low angle region. This is the clear indication of amorphous nature within the resolution limit of XRD instrument.



**Fig.1:** X-ray diffraction pattern of ZnLiBiB (HO) glasses.

### 4.2. Absorption spectra

The absorption spectra of ZnLiBiB (HO) glasses, consists of absorption bands corresponding to the absorptions from the ground state  $^5I_8$  of  $\text{Ho}^{3+}$  ions. Twelve absorption bands have been observed from the ground state  $^5I_8$  to excited states  $^5I_5$ ,  $^5I_4$ ,  $^5F_5$ ,  $^5F_4$ ,  $^5F_3$ ,  $^3K_8$ ,  $^5G_6$ ,  $(5G,3G)_5$ ,  $^5G_4$ ,  $^5G_2$ ,  $^5G_3$ , and  $^3F_4$  for  $\text{Ho}^{3+}$  doped ZnLiBiB(HO) glasses.



**Fig.2:** UV-VIS absorption spectra of ZnLiBiB (HO) glasses.

The experimental and calculated oscillator strengths for Ho<sup>3+</sup> ions in zinc lithium bismuth borate glasses are given in **Table 2**

**Table 2.** Measured and calculated oscillator strength ( $P^m \times 10^{+6}$ ) of Ho<sup>3+</sup> ions in ZnLiBiB glasses.

Energy level	Glass ZnLiBiB (HO01)		Glass ZnLiBiB (HO1.5)		Glass ZnLiBiB (HO02)	
	P <sub>exp.</sub>	P <sub>cal.</sub>	P <sub>exp.</sub>	P <sub>cal.</sub>	P <sub>exp.</sub>	P <sub>cal.</sub>
<sup>5</sup> I <sub>5</sub>	0.33	0.24	0.32	0.21	0.32	0.21
<sup>3</sup> I <sub>4</sub>	0.04	0.02	0.02	0.02	0.02	0.02
<sup>5</sup> F <sub>5</sub>	3.20	2.66	2.35	2.34	2.30	2.34
<sup>3</sup> F <sub>4</sub> , <sup>5</sup> S <sub>2</sub>	4.52	4.19	4.22	3.67	4.18	3.67
<sup>3</sup> F <sub>3</sub>	1.55	2.35	1.52	2.06	1.48	2.06
<sup>3</sup> K <sub>8</sub> , <sup>5</sup> F <sub>2</sub>	1.35	1.91	1.34	1.70	1.32	1.70
<sup>3</sup> G <sub>6</sub>	23.62	23.68	22.6	22.58	22.2	22.23
( <sup>3</sup> G, <sup>3</sup> G) <sub>5</sub>	3.34	1.52	3.32	1.35	3.20	1.34
<sup>3</sup> G <sub>4</sub> , <sup>3</sup> K <sub>7</sub>	0.07	0.58	0.05	0.52	0.03	0.52
<sup>3</sup> G <sub>2</sub> , <sup>3</sup> H <sub>5</sub>	5.21	5.08	4.40	4.77	4.10	4.71
<sup>3</sup> G <sub>3</sub> , <sup>3</sup> L <sub>9</sub>	1.40	1.35	1.30	1.20	1.10	1.20
<sup>3</sup> F <sub>4</sub> , <sup>3</sup> K <sub>6</sub>	1.32	3.82	1.28	3.39	1.22	3.38
R.m.s.deviation	0.967		0.887		0.889	

Computed values of F<sub>2</sub>, Lande's parameter ( $\zeta_{4f}$ ), Nephelauxetic ratio( $\beta'$ ) and bonding parameter ( $b^{1/2}$ ) for Ho<sup>3+</sup> doped ZnLiBiB glass specimen are given in **Table 3**.

**Table 3.** F<sub>2</sub>,  $\zeta_{4f}$ ,  $\beta'$  and  $b^{1/2}$  parameters for Holmium doped glass specimen.

Glass Specimen	F <sub>2</sub>	$\zeta_{4f}$	$\beta'$	$b^{1/2}$
Ho <sup>3+</sup>	427.89	2196.01	0.9718	0.1187

Judd-Ofelt intensity parameters  $\Omega_\lambda$  ( $\lambda = 2, 4, 6$ ) were calculated by using the fitting approximation of the experimental oscillator strengths to the calculated oscillator strengths with respect to their electric dipole contributions. In the present case the three  $\Omega_\lambda$  parameters follow the trend  $\Omega_4 < \Omega_6 < \Omega_2$ . The variation of  $\Omega_2$  with Bi<sub>2</sub>O<sub>3</sub> content has been attributed to changes in the asymmetry of the ligand field at the rare earth ion site and to the changes in their rare earth oxygen covalence.

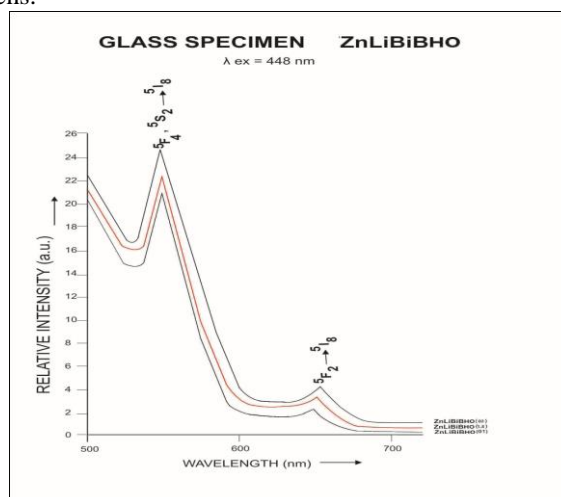
The values of Judd-Ofelt intensity parameters are given in **Table 4**.

**Table 4.** Judd-Ofelt intensity parameters for Ho<sup>3+</sup> doped ZnLiBiB glass specimens.

Glass Specimen	$\Omega_2(\text{pm}^2)$	$\Omega_4(\text{pm}^2)$	$\Omega_6(\text{pm}^2)$	$\Omega_4/\Omega_6$
ZnLiBiB HO01	4.942	1.021	1.828	0.559
ZnLiBiB HO1.5	4.776	0.910	1.593	0.571
ZnLiBiB HO02	4.686	0.904	1.597	0.566

#### 4.3. Fluorescence Spectrum

The fluorescence spectrum of Ho<sup>3+</sup> doped in zinc lithium bismuth borate glass is shown in Figure 3. There are two broad bands (<sup>5</sup>F<sub>4</sub>, <sup>5</sup>S<sub>2</sub> → <sup>5</sup>I<sub>8</sub>) and (<sup>5</sup>F<sub>5</sub> → <sup>5</sup>I<sub>8</sub>) respectively for glass specimens.



**Fig.3:** Fluorescence spectrum of ZnLiBiB glasses doped with Ho<sup>3+</sup>.

**Table 5.** Emission peak wave lengths ( $\lambda_p$ ), radiative transition probability ( $A_{rad}$ ), branching ratio ( $\beta_R$ ), stimulated emission crosssection ( $\sigma_p$ ), and radiative life time ( $\tau_R$ ) for various transitions in Ho<sup>3+</sup> doped ZnLiBiB glasses.

Transition	ZnLiBiB(HO01)					ZnLiBiB(HO1.5)				ZnLiBiB(HO02)			
	$\lambda_p$ (nm)	$A_{rad}(s^{-1})$	$\beta_R$	$\sigma_p$ (10 <sup>-20</sup> cm <sup>2</sup> )	$\tau_R$ ( $\mu$ s)	$A_{rad}(s^{-1})$	$\beta_R$	$\sigma_p$ (10 <sup>-20</sup> cm <sup>2</sup> )	$\tau_R$ ( $\mu$ s)	$A_{rad}(s^{-1})$	$\beta_R$	$\sigma_p$ (10 <sup>-20</sup> cm <sup>2</sup> )	$\tau_R$ ( $\mu$ s)
<sup>3</sup> F <sub>4</sub> <sup>5</sup> S <sub>2</sub> → <sup>3</sup> I <sub>8</sub>	555	5888.75	0.7241	1.19	122.96	5163.03	0.7233	0.97	140.09	5176.97	0.7237	0.95	139.79
<sup>3</sup> F <sub>3</sub> → <sup>3</sup> I <sub>8</sub>	652	2244.15	0.2759	1.09		1975.31	0.2767	0.92		1976.80	0.2763	0.89	

## V. Conclusion

In the present study, the glass samples of composition (25-x) Bi<sub>2</sub>O<sub>3</sub>:20Li<sub>2</sub>O: 20ZnO:35B<sub>2</sub>O<sub>3</sub>:xHo<sub>2</sub>O<sub>3</sub> (where x =1, 1.5, 2 mol %) have been prepared by melt-quenching method. The value of stimulated emission cross-section ( $\sigma_p$ ) is found to be maximum for the transition (<sup>3</sup>F<sub>4</sub>, <sup>5</sup>S<sub>2</sub>→<sup>3</sup>I<sub>8</sub>) for glass ZnLiBiB (HO 01), suggesting that glass ZnLiBiB (HO 01) is better compared to the other two glass systems ZnLiBiB (HO1.5) and ZnLiBiB (HO02).

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